

# Can cluster physics shed some light on the puzzles of our Experimental findings with energetic particles?

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## Introduction

Leakage current /  
Carrier removal / ???clusters??  
Annealing /

## 1. Cluster formation

(Vinetskii and Kondrachuk in Rad. Effects 1975, vol.30 p227)

first stage: high-energy particles displace matrix atoms  
point defects: vacancies(V), interstitials(I)

second stage: diffusion of vacancies and interstitials  
quasimolecules (secondary radiation defects)

### Model for second stage:

Mobility of  $I \gg V$

2 reactions  $V+V \rightarrow D$  reaction constants  
 $V+O \rightarrow A$   $\alpha_D, \alpha_A$

Diffusion equation for vacancies and reaction equation  
for Oxygen and initial conditions at center of cluster  $N_0$   
system of nonlinear equations  
dimensionless parameter

$$\mathcal{E} = \frac{D_V}{\alpha_D N_0 L^2}$$

L characteristic size of cluster

N and L depend on incident particle, while  $D_V, \alpha_D$  and

Depend on crystal properties

- a)  $\mathcal{E} \ll 1$  low vacancy diffusion  $\rightarrow$  congealing cluster  
shape of D-cluster  $\sim$  V-cluster (original)
- b)  $\mathcal{E} \gg 1$  high vacancy diffusion  $\rightarrow$  spreading cluster

belt of A centers around D-cluster

energy threshold of knock-on atom between 2 cases 10keV

## 2. Properties of cluster

2.1 structure  $\rightarrow$  central region (core): vacancies + interstitial associations  
 $\searrow$  peripheral region (impurity defect shell)  
 bell shape of distribution

2.2 electrical field center: charged D depending on Fermi level  
 periphery: charged impurity defects  
 local charge neutrality: (acceptors, donors, free carriers)

Poisson equation + charge neutrality  $\rightarrow$  selfconsistent system  
 which yields occupation probability of acceptors  $f_i$   
 and potential barrier  $\psi_0$  between matrix and center of cluster  
 Approximate solution of Poisson equation

$$\psi_0 \approx \frac{2\pi e^2 r^2}{\epsilon} \sum_i N_i f_i - \frac{kT}{3}$$

$$f_i = \frac{1}{1 + N_c/n_g \exp \left[ \frac{\psi_0 - \Delta E_i}{kT} \right]}$$

where  $r$  is radius of cluster and  $\Delta E_i$  energy position of defect in forbidden bandgap  
 $N_i$  density of defect

### 2.3 Parameter description

Kuznetsov and Lugakov, phys.stat.sol(a) vol.79, p381 (1983)  
Temperature dependencies of Hall coefficient + nature of defect

Defect cluster parameters in n-Si

irradiation	material	$n_0$ ( $10^{13} \text{ cm}^{-3}$ )	core			periphery		
			$\psi$ (eV)	radius (Å)	radiation defect density ( $\text{cm}^{-3}$ )	$\psi$ (eV)	radius (Å)	radiation defect density ( $\text{cm}^{-3}$ )
neutrons	float-zone	3	$\approx 0.14$	$\approx 130$	$\approx 8 \times 10^{17}$	$\approx 0.03$	$\approx 600$	$\approx 1.7 \times 10^{16}$
neutrons	pulled	2	$\approx 0.14$	$\approx 130$	$\approx 8 \times 10^{17}$	0.05	300	$1.2 \times 10^{17}$
protons	float-zone	1	0.11	1800	$7 \times 10^{15}$	0.02	15000	$2.7 \times 10^{13}$
protons	float-zone	3	0.11	1800	$7 \times 10^{15}$	0.023	13000	$4.3 \times 10^{13}$
protons	float-zone	8	0.11	1800	$7 \times 10^{15}$	0.025	12000	$5.8 \times 10^{13}$
protons	float-zone	1.6	0.11	1800	$7 \times 10^{15}$	0.03	9000	$1.6 \times 10^{14}$
protons	pulled	2	0.11	1800	$7 \times 10^{15}$	0.03	6000	$2.8 \times 10^{14}$
protons	pulled	5	0.11	1800	$7 \times 10^{15}$	0.033	6000	$2.8 \times 10^{14}$
protons	pulled	30	0.11	1800	$7 \times 10^{15}$	0.04	5000	$5.3 \times 10^{14}$

### 2.4 Strain field

Deposited energy ~10000 eV in  
a cluster volume ~10 E-18 cm-3  
Temperatur diff ~1000 K  
Fast cooling in ~10E-10 s  
Pressure ~10000 bar

### 2.5 Divacancy levels in cluster

A.V. Vasil'ev et al. Sov.Phys.Semicond. vol.20(4) p465  
(1986)

For locally inhomogeneous distribution of deep levels

$$N_d^- = N_r \int_V F(r) f(T, E_F - E_d) dV$$

where  $\int_V F(r) dV = 1$

and  $f(T, E_F - E_d)$  occupancy

$N_d^-$  Volume average concentration of charged center

$N_r$  Concentration of local region

Analysis performed with following assumptions

- homogeneous distribution of D in matrix + embedded local regions with M divacancies in it.
- bell shape of initial cluster

$$F(r) = \frac{1}{\pi^{3/2} L^3} \exp(-r^2/L^2)$$

- only divacancy level  $E_c - 0.39$  eV considered
- N depends linearly on dose

A. V. Vankov et al.

Sov. Phys. Semicond.

20 p 465 (1986)

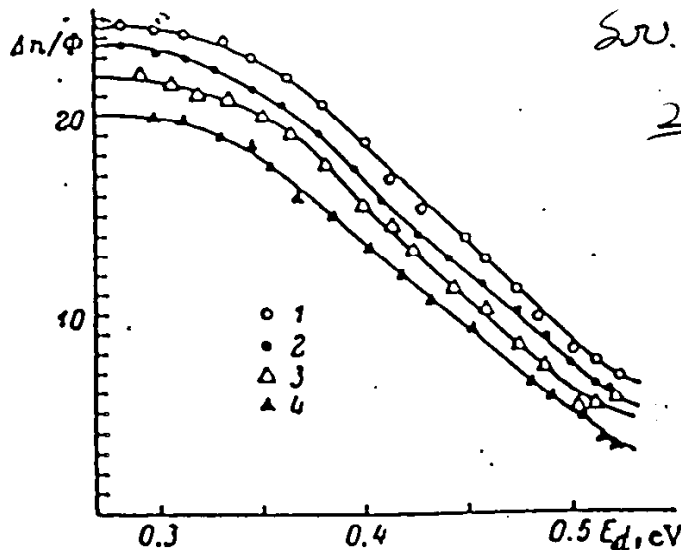


FIG. 1. Values of  $\Delta n/\Phi$  obtained at temperatures 250-450 K for zone-grown (1) and crucible-grown (2-4) samples. Radiation dose  $\Phi$  ( $10^{12} \text{ cm}^{-2}$ ): 1), 2) 0.5; 3) 1.0; 4) 1.5. The experimental points were taken from Ref. 7. The continuous curves are theoretical dependences calculated on the assumption of a locally inhomogeneous distribution of divacancies in a crystal (Table II).

$$n_0 = 5 \cdot 10^{13}$$

$\alpha$  at 6.3 MeV

no  $V = \text{stef}!!$

### 3. Consequences for ROSE problems

#### 3.1 Carrier Recombination

Most likely recombination centers are D

Problem of capture coefficient for free carriers

If there is a potential barrier built-up inside cluster:

Holes attracted into interior

Electrons pushed to matrix depletion

Then effective capture cross section for holes

$$\sigma_p^* = \sigma_p \exp\left(\frac{\psi}{kT}\right) \cdot K_s$$

$K_s$  is a factor for fact that potential barrier  $\psi$  is not abrupt but extends to a Debeye screening length for the case that this length is smaller than outer cluster radius

$K_s \sim \text{vol of total cluster} / \text{volume of negativ charge}$

For  $n_0 \approx 3 \cdot 10^{13} \text{ cm}^{-3}$   $K_s \exp\left(\frac{\psi}{kT}\right) \sim 30$

P.F.Lugakov and V.V.Shusha Rad. Effects vol.62 p197 (1982)  
and Stefanov et al. Phys.stat. sol.(a) vol 163 p27 (1997)  
find

$\sigma_p / \sigma_n \approx 150$  for clusters

and  $\sigma_p / \sigma_n \approx 22$  electron irradiated Si

This yields  $K_s \exp\left(\frac{\psi}{kT}\right) \approx 7$  for 2kOhmcm Si

M.Moll needs 6

### 3.2 DLTS measurements

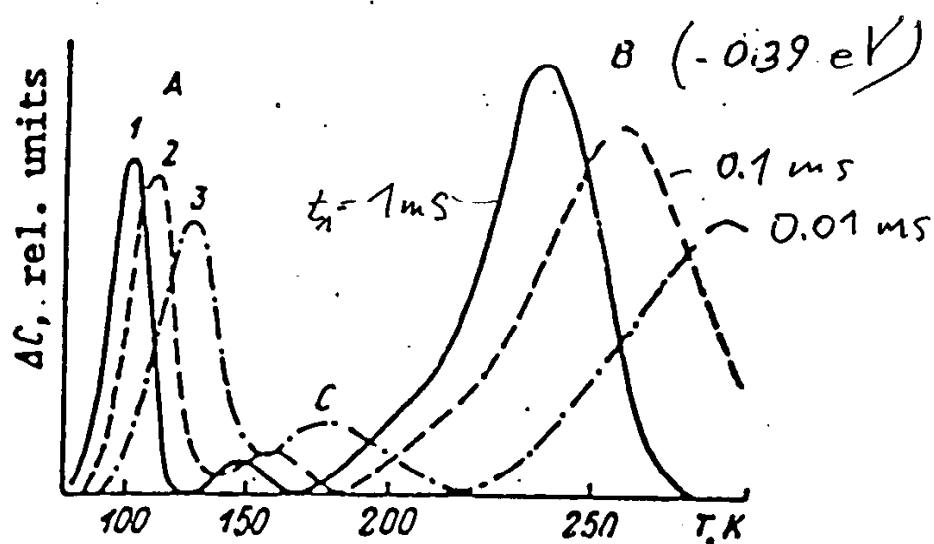
Careful with interpretation, since charging of traps in potential well of clusters not guaranteed

Capture rate of carriers  $\gg$  thermal release of carriers

I.V. Antonova et al. Sov.Phys.Semicond. vol.22(6) p630 (1988)

Broadening + temperature shift of E peak

reactor neutrons



and Ch.M.Hardalov et al. Appl.Phys. A61 p107 (1995)

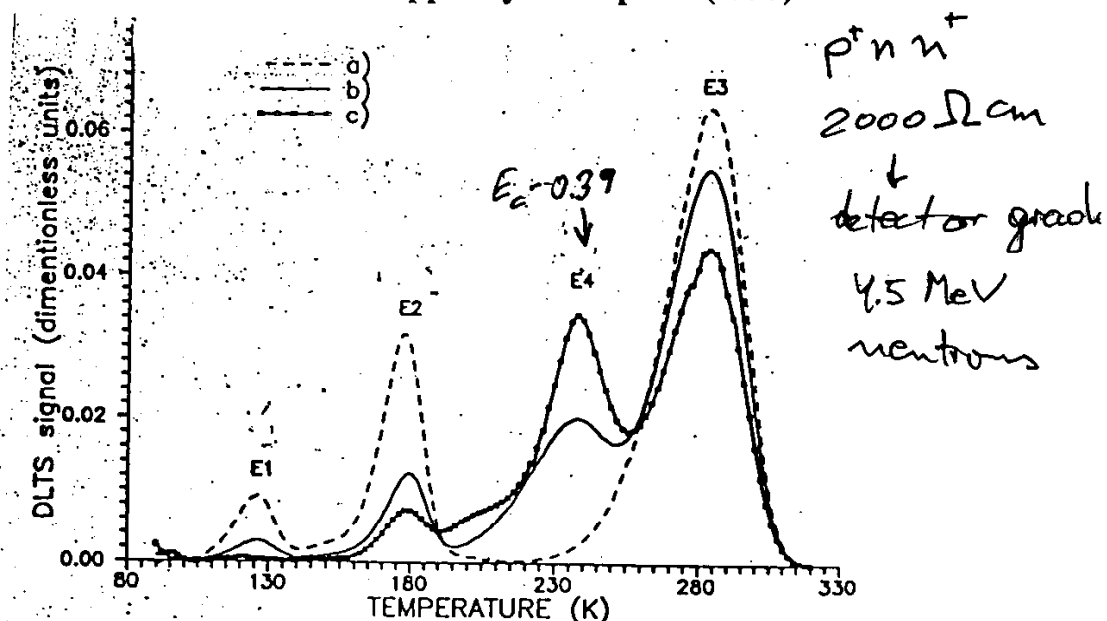


Fig. 1. DLTS spectra of non-irradiated sample (a); irradiated with a fluence of  $5.5 \times 10^{11} \text{ n/cm}^2$  (b); and with a fluence of  $1.0 \times 10^{12} \text{ n/cm}^2$  (c)

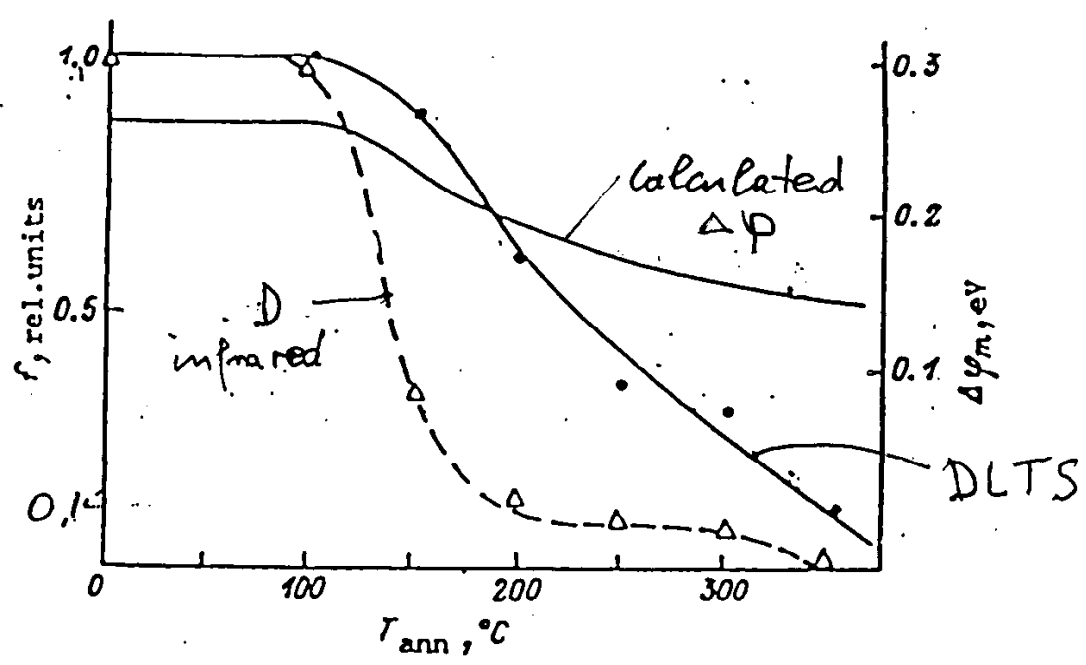
	E1	E2	E3	E4	$\Phi_e \text{ cm}^{-2}$
Before radiation	0.21	0.31	0.49		
After $5.5 \cdot 10^{11}$				0.43	$2 \cdot 10^{-15}$
After $10 \cdot 10^{12}$				0.33	$2 \cdot 10^{-17}$

Extend of cluster space charge layer (CSCL) ~ 3 microns  
Debye length

Overlap of CSCL after a fluence  $10 \cdot 10^{12} \text{ cm}^{-2}$   
Fabrication defects (shallow) masked by local potential

### 3.3 Annealing of D in Si containing clusters

3.31 I.V.Antonova et al. Sov.Phys.Semicond. vol .23 p671 (1989)



$CZ \text{ nt } 3.5 \cdot 10^{15}$   
 $\Phi \ 4.4 \cdot 10^{-14} \text{ cm}^{-2}$

$D = 95$   
 $R_0 = 220 \text{ \AA}$

observation: anneal in 2 stages

250 - 350°C classical  $E_a = 1.5\text{eV}$

100 - 200°C smooth / different mechanism

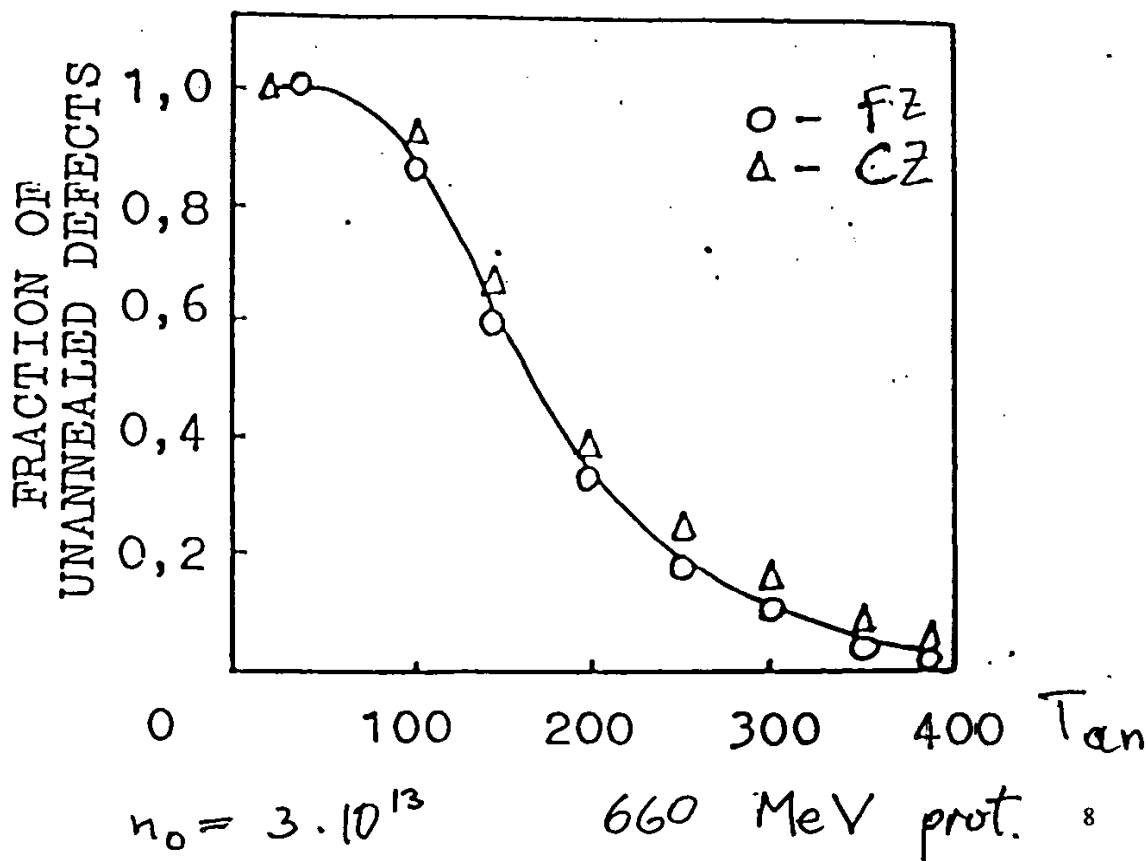
at 120°C  $I_2 \longrightarrow I + I$

$I + C_S \longrightarrow C_i$

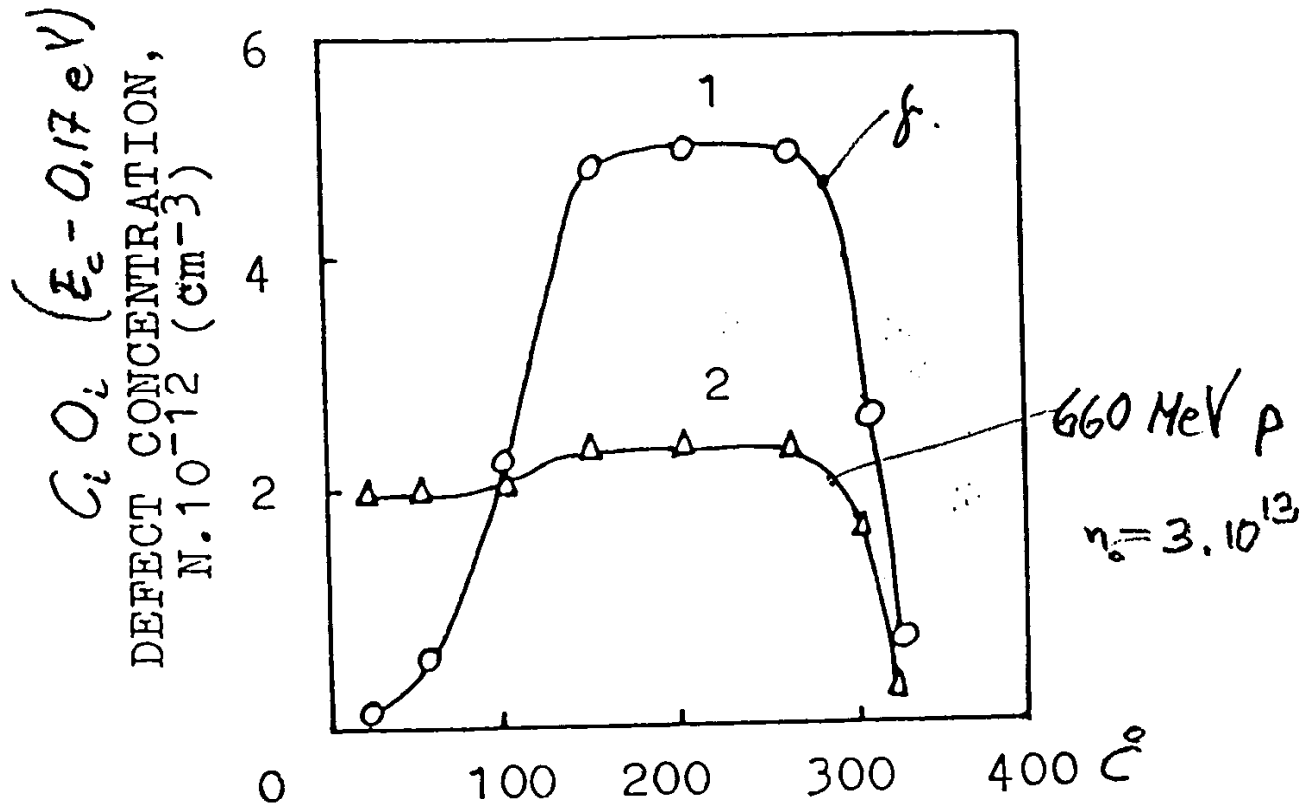
$C_i \longrightarrow \text{cluster (strain)}$

### 3.31 Peculiar observations

V.I. Kuznetsov et al. Rad. Effects Lett. Vol.86 p199 (1984)





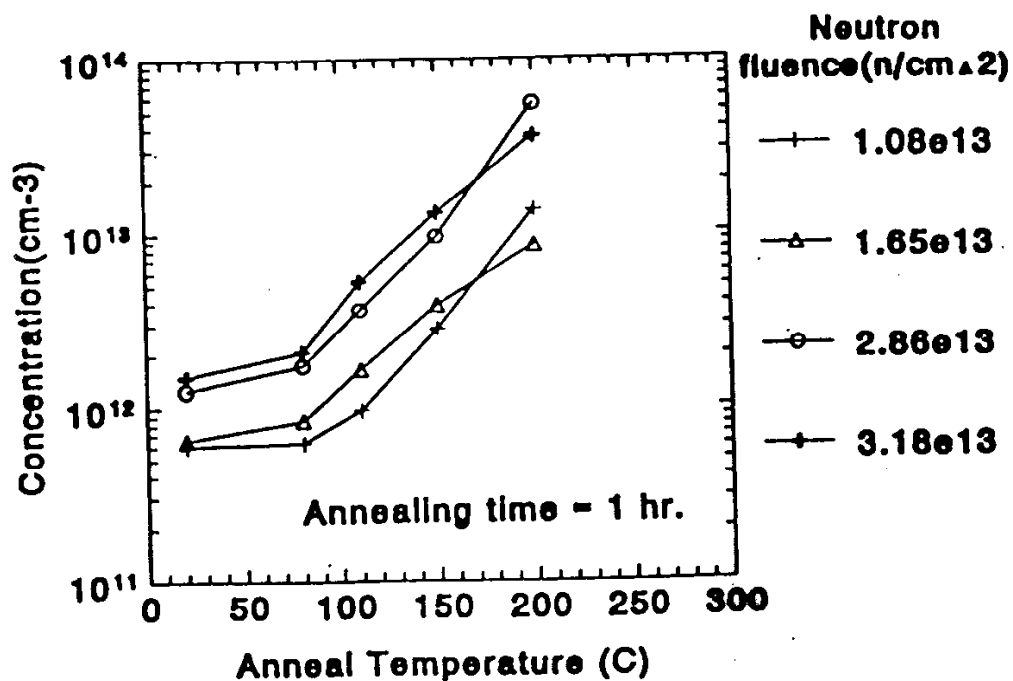


Indirect proof of  $C_i$  annealing: more than 2 times of  $C_i$  goes into cluster

Effect of strain field:  $C_i$  cause lattice compression  
 $D$  cause lattice tension

### 3.3 Annealing of ROSE detectors

Z.Li et al. NIM A 385 p321 (1997)

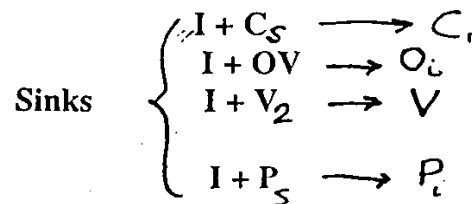


#### 4. Model for positive and negative annealing

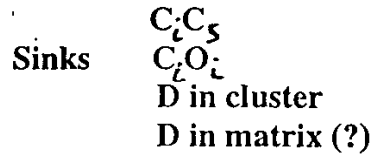
Be cautious!! Might be crab!!

Suppose: main defect D distributed as point defects in matrix  
or forming a cluster  
potential barriers and strain field around cluster

irradiation produces Interstitials which go to sinks



$C_i$  are mobile and get eventually settled in



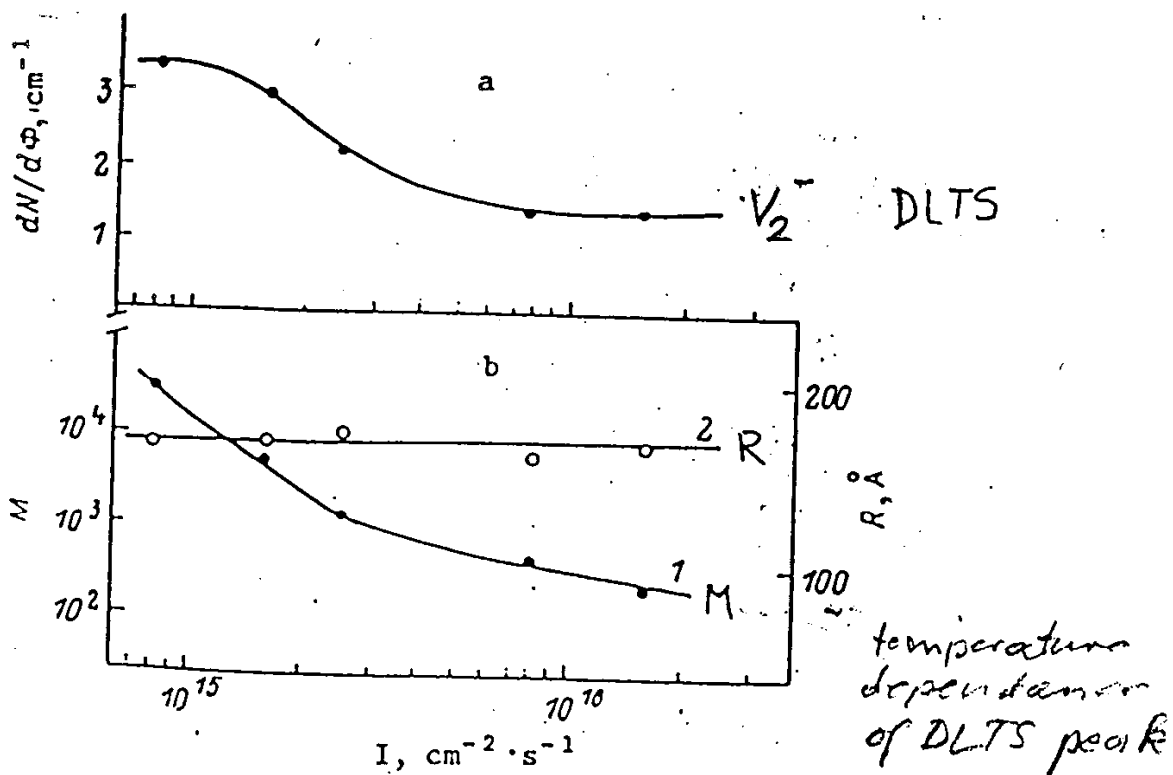
4.1 Positive annealing: destruction of D in vacancies by C

4.2 Negative annealing : destruction of D in clusters by C

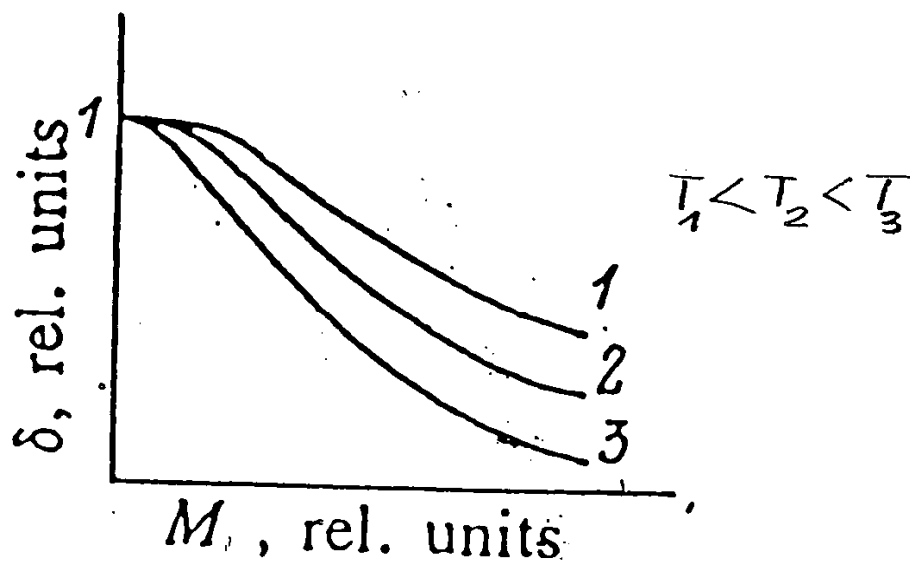
Why??

Because destruction of D lowers potential barrier and more  
Charged D become visible

I.V. Antonova et al. Sov. Phys. Semicond. vol.23 p 944 (1989)



I.V. Antonova and S.S. Shaimeev, Sov. Phys. Semicond. vol.25 p513 (1991)



I.V. Antonova et al. Sov. Phys. Semicond. vol.22 p630 (1988)

have shown that  $\psi_{max} \sim \ln D$

if D goes down D goes up

since  $\Delta \psi_{max} \sim \frac{\Delta D}{D}$

$\Delta \psi_{max}$  changes strongly when few D left

this is reason why negative annealing delayed

#### 4.3 Some experimental facts in favor of model

4.31 decrease of overall D measured by positron lifetime and infrared technique during annealing

4.32 Thesis M. Moll (1999) p198

peak H(116K)  $\uparrow$  assignment to  $C_i^{+1/0}$   
because C pushed in by strain field and gets immobile

appearance of small peak  $V_2^=$

$C_i O_i$  and probably  $C_i C_S$  go up

Unexplained time behavior of reverse annealing

#### 5. Conclusion and Outlook

some properties of clusters are able to explain peculiar behavior of our measurements; it has to be proven that clusters produced in ROSE detectors behave as the ones found in low resistivity material

in particular geometrical complications are expected due to the great Debye length in ROSE detectors. (merging of the periphery of clusters)

**qualitative picture: further thinking if correct; refine or amend**

**quantitative model : needs description of primary cluster with  
input from experiment ( probably dedicated)  
since cluster shape depends also on chemical impurities whole kinetic of  
V and I- diffusion must be implemented  
Calculate cluster size and number of D  
Do point defect kinetics in periphery of cluster**